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Introduction

As the light neighbour of carbon in the periodic table with prototypical electron-deficiency, boron exhibits unique structures and bonding in its polyhedral molecules, bulk allotropes,¹ and the theoretically predicted buckyballs² and newly observed bilayer borophenes.³ In the past two decades, persistent joint photoelectron spectroscopy (PES) and first-principles theory investigations have revealed that small boron clusters B_n⁻ possess planar or quasi-planar (n = 3-38, 41, and 42), cagelike $(B_{40}^{-/0}/B_{39}^{-})$, and bilayer $(B_{48}^{-/0})$ structures in a wide size range between n = 3-48, featuring both delocalized σ and π multicenter-two-electron (*m*c-2e) bonding interactions.⁴⁻⁹ Combined ion-mobility measurements and density functional theory (DFT) investigations indicated that cationic boron clusters B_n^+ possess double-ring tubular structures (n = 16-25).¹⁰ The experimentally observed cage-like $B_{40}^{-/0}/B_{39}^{-}$ have been extended to the B_n^q borospherene family (n = 36-42, q = n - 40)

Lanthanide/actinide boride nanoclusters and

nanomaterials based on boron frameworks

consisting of conjoined B_n rings (n = 7-9)

in this work indicate that the experimentally observed perfect inverse sandwich lanthanide boride complexes $D_{7h} La_2B_7^-$ (**1**), $D_{8h} La_2B_8$ (**3**), $D_{9h} La_2B_9^-$ (**7**) can be extended to their actinide counterparts $C_{2v} Ac_2B_7^-$ (**1**'), $D_{8h} Ac_2B_8$ (**3**'), $D_{9h} Ac_2B_9^-$ (**7**') with a B_n monocyclic ring (n = 7-9) sandwiched by two Ac dopants. Such $M_2B_n^{-/0}$ inverse sandwiches (**11**', **313**', **77**') can be used as building blocks to generate the ground-state $C_2 La_4B_{13}^-$ (**2**)/Ac_4B_{13}^- (**2**'), $D_2 La_4B_{15}^-$ (**4**)/Ac_4B_{15}^- (**4**'), $C_{3v}/C_3 La_4B_{18}$ (**5**)/Ac_4B_{18} (**5**'), $O_h Ac_7B_{24}^+$ (**6**'), $O_h Ac_7B_{24}^+$ (**6**'), $O_h Ac_7B_{24}^+$ (**6**'), $O_h Ac_7B_{24}^-$ (**10**') which are based on boron frameworks consisting of multiple conjoined B_n rings (n = 7-9). Detailed bonding analyses show that effective $(d-p)\sigma$, $(d-p)\pi$ and $(d-p)\delta$ coordination bonds are formed between the B_n rings and metal doping centers, conferring three-dimensional aromaticity and extra stability to the systems. In particular, the perfect body-centered cubic $O_h Ac_7B_{24}^+$ (**6**') and $O_h Ac_7B_{24}$ with six conjoined B₈ rings can be extended in *x*, *y*, and *z* dimensions to form one-dimensional $Ac_{10}B_{32}$ (**11**'), two-dimensional Ac_3B_{10} (**12**'), and three-dimensional Ac_6 (**13**') nanomaterials, presenting a B₈-based bottom-up approach from metal boride nanoclusters to their low-dimensional nanomaterials.

at first-principles theory level, while the bilayer structural motif firstly observed in ${\rm B_{48}}^{-/0}$ has been expanded recently to a series of novel bilayer boron clusters in the size range between ${\rm B_{50}-B_{72}}$ and a bottom-up approach from medium-sized boron clusters to bilayer borophenes at DFT level.¹¹ In addition, icosahedral-B₁₂ stuffed B₈₀, B₈₄, B₁₀₂, and B₁₁₂ have been predicted at DFT level, with $C_{\rm s}$ B₁₁₂ as the most stable coreshell borospherene with one B₁₂ icosahral core at the center reported to date.¹²⁻¹⁵ The structures of small and, especially, medium-sized boron clusters appear to exhibit strong dependence on both their sizes and charge states.

Transition-metal-doping has proven to induce dramatic structural changes to boron clusters, greatly enriching the structural diversity of metal boride complexes. Mono-metal-doped boron clusters exhibit metal-centered planar wheel-like, half-sandwich, or tubular drum-like structures, generating the coordination number of CN = 10 in planar D_{10h} Ta \bigcirc B₁₀⁻ and D_{10h} Nb \bigcirc B₁₀⁻¹⁶⁻²¹ A recent investigation indicates that the experimentally observed perfect planar D_{13h} La \bigcirc C₁₃⁺ possesses the record coordination number of CN = 13 in planar species.²² A series of perfect inverse-sandwich di-metal-doped boron clusters D_{6h} Ta₂B₆^{-/0}, D_{7h} La₂B₇⁻ (1), D_{8h} La₂B₈ (3), and D_{9h} La₂B₉⁻ (7) were recently discovered in PES measurements, with a perfect planar B_n ring (n = 6-9) sandwiched between two metal atoms *via* effective (d-p) σ , (d-p) π , and (d-p) δ coordination



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Xiao-Qin Lu,^{ab} Xiao-Ni Zhao,^a Yue-Wen Mu^(b) *^a and Si-Dian Li^(b) *^a Extensive global minimum searches augmented with first-principles theory calculations performed in this work indicate that the experimentally observed perfect inverse sandwich lanthanide boride complexes D_{7h} La₂B₇⁻ (**1**), D_{8h} La₂B₈ (**3**), D_{9h} La₂B₉⁻ (**7**) can be extended to their actinide counterparts

Paper

bonding interactions.²³⁻²⁵ However, both $C_1 \text{ La}_2 \text{B}_{10}^-$ and C_{2v} La₂B₁₁⁻ were experimentally characterized to be expanded inverse sandwiches based on a B₈ ring, indicating that the B₉ is the largest boron ring to form perfect inverse-sandwich metal boride complexes.²⁶ The tri-La-doped inverse triple-decker C_{2v} $La_3B_{14}^{-}$ was discovered in PES experiments with two conjoined B₈ rings sharing a B-B edge.²⁷ PES experiments indicate $La_{3}B_{18}^{-}$ is a perfect D_{3h} trihedral metallo-borospherene with three deca-coordinate La atoms on the cage surface.²⁸ Such a metallo-borospherene was late theoretically extended to the smallest core-shell metallo-borospherene D_{3h} La₃B₂₀ $(B_2(a)La_3B_{18})$ with a B_2 dumb-bell at the cage center along the C₃ molecular axis.²⁹ Based on extensive global minimum (GM) searches and first-principles theory calculations, our group recently proposed a series of novel lanthanide boride nanoclusters including the smallest double-ring tubular molecular rotor C_{2h} La₂[B₂@B₁₈] with a B₂ dumb-bell inside,³⁰ perfect endohedral cubic $O_h \operatorname{La}_7 \operatorname{B}_{24}^+$ (6) and $O_h \operatorname{La}_7 \operatorname{B}_{24}$ with six conjoined inverse La2B8 sandwiches sharing a La atom at the center, perfect tetrahedral metallo-borospherene T_{d} La₄B₂₄ (8) with four nona-coordinate La atoms on the cage surface, perfect tetrahedral core-shell metallo-borospherene $T_{\rm d} \operatorname{La}_4 \operatorname{B}_{29}^{0/+/-}$ (10) (B@B₄@La₄B₂₄) with a tetrahedral BB₄ unit at the center, and endohedral trihedral metallo-borospherenes D_{3h} La@[La₅&B₃₀]^{0/2-} with spherical aromaticity.³¹⁻³³ However, tetra-La-doped boride complexes $La_4B_n^{-/0}$ (n = 13-18) with conjoined B7, B8, and B9 rings as ligands still remain unknown in both experiments and theory, and more importantly, the chemistry and materials science of the corresponding actinide boride nanoclusters and their low-dimensional nanomaterials have not been explored to date in either experiments or theory possibly due to the radioactivity of actinide. Modern quantum chemistry provides powerful means to fulfil the gaps in such situations.

Inspired by the experimentally observed perfect inverse sandwich $La_2B_7^-$ (1), La_2B_8 (3), $La_2B_9^-$ (7) and based on extensive first-principles theory calculations, we predict herein a series of novel La/Ac doped boron clusters including C_{2v} $Ac_2B_7^-$ (1'), D_{8h} Ac_2B_8 (3'), D_{9h} $Ac_2B_9^-$ (7'), C_2 $La_4B_{13}^-$ (2)/ $Ac_4B_{13}^-$ (2'), D_2 $La_4B_{15}^-$ (4)/ $Ac_4B_{15}^-$ (4'), C_{3v}/C_3 La_4B_{18} (5)/ Ac_4B_{18} (5'), O_h $Ac_7B_{24}^+$ (6'), T_d Ac_4B_{24} (8'), C_1 $La_5B_{24}^+$ (9)/ $Ac_5B_{24}^+$ (9'), and T_d $Ac_4B_{29}^-$ (10') which are composed of mono or multiple conjoined B_n rings (n = 7-9) as ligands to coordinate the metal centres. More interestingly, the cubic $Ac_7B_{24}^{+/0}$ (6') can be extended to form the highly stable actinide boride nanomaterials 1D $Ac_{10}B_{32}$ (11'), 2D Ac_3B_{10} (12'), and 3D AcB_6 (13'), expanding the previously reported low-dimensional lanthanide boride nanomaterials to their actinide boride counterparts.

Theoretical methods

The global-minimum (GM) structures of $La_4B_{13}^{-}$, $La_4B_{15}^{-}$, $La_4B_{18}^{-}$, La_4B_{1

based on the experimentally observed La-B binary complexes. More than 3000 trial structures were explored for each species at PBE/TZVP. The low-lying isomers were subsequently optimized using the hybrid PBE0³⁵ and TPSSh³⁶ functional with the 6-311+G* basis set³⁷ for B and Stuttgart energy-consistent relativistic pseudopotential ECP46MWB38,39 and ECP78MWB40 for La and Ac using the Gaussian 09 program.41 All local minima were checked by vibrational frequency analyses. To obtain more accurate relative energies, we did further single-point CCSD(T) calculations⁴²⁻⁴⁴ at the PBE0 geometries with the same basis sets for the five lowest-lying isomers of La₄B₁₃⁻, La₄B₁₅⁻, and La₄B₁₈ using the Molpro package.45 Based on the obtained low-lying isomers of La₂B₇⁻, La₄B₁₃⁻, La₂B₈, La₄B₁₅⁻, La₄B₁₈, La₇B₂₄⁺, $La_2B_9^{-}$, La_4B_{24} , $La_5B_{24}^{+}$, and $La_4B_{29}^{-}$, we optimized the structures of their actinide boride counterparts Ac2B7, Ac4B13, Ac2B8, Ac₄B₁₅⁻, Ac₄B₁₈, Ac₇B₂₄⁺, Ac₂B₉⁻, Ac₄B₂₄, Ac₅B₂₄⁺, and Ac₄B₂₉⁻. Low dimensional actinide boride nanostructures 1D $Ac_{10}B_{32}$, 2D Ac₃B₁₀, and 3D AcB₆ constructed from cubic Ac₇B₂₄⁺ (6') were fully optimized using the Vienna ab initio simulation package (VASP),^{46,47} within the framework of projector augmented wave (PAW)^{48,49} pseudopotential method and PBE generalized gradient approximation (GGA).⁵⁰ The Coulomb-corrected local spin-density approximation (LSDA + U) was utilized for both structural relaxation and static calculation (U = 5 eV) for La and Ac.^{51,52} Different long-range corrections with U = 3, 4, and 5 are found to produce essentially the same lattice constants and band structures for lowdimensional actinide materials. The cutoff energy of 500 eV was used. The criterion of convergence for the maximum force on each atom was less than 0.01 eV $Å^{-1}$ and the total energy was 10^{-5} eV. Chemical bonding was analyzed using the adaptive natural density partitioning (AdNDP) approach.⁵³ Natural bonding orbital (NBO) analyses were performed using the NBO 6.0 program.⁵⁴ Born–Oppenheimer molecular dynamics (BOMD) simulations were carried out on $La_4B_{13}^{-}$ (2), $La_4B_{15}^{-}$ (4), and La_4B_{18} (5) for 30 ps at 800 K using the CP2K code.⁵⁵ The iso-chemical shielding surfaces (ICSSs)⁵⁶ were generated with the Multiwfn 3.8 code.57

Results and discussion

Lanthanide/actinide boride nanoclusters consisting of conjoined B_n rings (n = 7-9)

As elements in the same group in the periodic table, $Ac([Rn]7s^26d^1)$ and $La([Xe]6s^25d^1)$ with similar valence electronic configuraitons are expected to form similar boroide nanoclusters and nanomaterials. It turned out to be true that, by substitution of La with Ac, the experimentally observed inverse sandwich lanthanide boride clusters D_{7h} $La_2B_7^{-/0}$ (1, ${}^{3}A_2'$), D_{8h} La_2B_8 (3, ${}^{3}A_{2g}$), D_{9h} $La_2B_9^{-}$ (7, ${}^{1}A_1'$) can be readily extended to their actinide counterparts D_{7h} $Ac_2B_7^{-}$ (${}^{3}A_2'$), D_{8h} Ac_2B_8 (3', ${}^{3}A_{2g}$), and D_{9h} $Ac_2B_9^{-}$ (7', ${}^{1}A_1'$) at PBE0, with the average optimized B–B, Ac–B, and Ac–Ac distances of $r_{B-B} = 1.587$, 1.554, and 1.542 Å, $r_{Ac-B} = 2.705$, 2.807, and 2.890 Å, and $r_{Ac-Ac} = 3.987$, 3.878, and 3.617 Å, respectively. However, at the more accurate CCSD(T) level, the triplet D_{7h} $Ac_2B_7^{-}$ is slightly

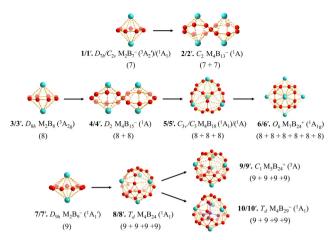


Fig. 1 Lanthanide and actinide boride nanoclusters consisting of mono or multiple conjoined B_n rings (n = 7-9) (M = La, Ac). The number of B_n rings (n = 7-9) in each structure is indicated in parentheses.

distorted to singlet $C_{2v} \operatorname{Ac_2B_7}^-(\mathbf{1}', {}^{1}A_1)$ which appears to be 0.56 eV more stable than its D_{7h} counterpart (Fig. S1(a), ESI†). $\operatorname{Ac_2B_n}^{-/0}(\mathbf{1}', \mathbf{3}', 7')$ as the global minima of the systems appear to lie at least 0.56 eV lower than their counterparts in other structural motifs at CCSD(T) (Fig. S1–S3, ESI†).

Using inverse sandwich $M_2B_7^{-}(1/1')$ as building blocks, the axially chiral $C_2 M_4 B_{13}^{-} (2/2')$ based on a B_{13} boron framework consisting of two conjoined B7 rings sharing a periphery B atom at the center in a ∞ -shaped structural motif (7 + 7 mode) were obtained. Interestingly and encouragingly, as shown in Fig. 1 and Fig. S1(2) (ESI⁺), extensive GM searches and first-principles calculations indicate that $C_2 \operatorname{La}_4 B_{13}^{-}$ (2, ¹A) composed of two conjoined $La_2B_7^{-}(1)$ inverse sandwiches is the well-defined GM of the system with the lowest vibrational frequency of v_{min} = 79.3 cm⁻¹ which lies 0.82 eV lower in energy than its triplet counterpart $D_2 \operatorname{La}_4 \operatorname{B}_{13}^-$ (³B₂) at CCSD(T), with all other lowlying isomers lying at least 0.41 eV higher in relative energy. In addition to the shared B atom, $La_4B_{13}^{-}$ (2) also forms two vertex-sharing B₃ triangles involving the central B and its four closest periphery boron neighbors as bridges to link the two equivalent B₇ rings together, faciliating the formation of multiple-center bonding interactions over the two vertexsharing B₃ triangles to help stabilize the system as detailed below. As shown in Fig. S4 (ESI^{\dagger}), La₄B₁₃⁻ (2) possesses the optimized B–B distance of r_{B-B} = 1.75 Å between central B and its four closest neighbors which is longer than the average B-B distance of $r_{\rm B-B}$ = 1.62 Å in the remaining parts of the two conjoined B₇ rings, while the La–B distances of $r_{\text{La–B}} = 2.63$ Å between the central B and four La atoms appear to be shorter than the remaining La–B distances with $r_{\text{La-B}} = 2.68-2.71$ Å. Detailed NBO analyses on $La_4B_{13}^{-}(2)$ show that central B atom possesses the natural atomic charge of $q_{\rm B} = -0.63$ |e| and total Wiberg bond order of $WBI_B = 3.92$, while the four equivalent La atoms carry the natural atomic charge of $q_{\text{La}} = +1.10$ |e| and total Wiberg bond order of $WBI_{La} = 3.27$. The B-B bond order between central B and its four neighboring B is calculated to be WBI_{B-B} = 0.60. Molecular dynamics simulations in 30 ps

indicate that $La_4B_{13}^{-}$ (2) is highly dynamically stable, with the small calculated average root-mean-square-deviations of RMSD = 0.13 Å and maximum bond length deviations of MAXD = 0.37 Å at 800 K (Fig. S5a, ESI†). Similar to $La_4B_{13}^{-}$ (2), $C_2 \ Ac_4B_{13}^{-}$ (2', ¹A) also proves to be the GM of the system (Fig. S1(2') (ESI†)) dynamically table at 800 K. (Fig. S5b, ESI†).

Employing M_2B_8 (3/3') as building blocks, we obtained the axially chiral singlet D_2 M₄B₁₅⁻ (4/4') which possesses two conjoined equavalent B_8 rings in a ∞ -shaped strucutral motif (8 + 8 mode), with two vertex-sharing B_3 triangles formed between the B atom at the center and its four closest boron neighbors, similar to the situation in $M_4B_{13}^{-}$ (2/2'). D_2 La₄ B_{15}^{-} (4) proves to be the well-defined GM of the system (Fig. S2, ESI⁺) with the lowest vibrational frequency of $v_{min} = 65.5 \text{ cm}^{-1}$ at PBE0 which lies 0.75 and 0.79 eV lower at CCSD(T) than the second (C_1) and third (C_s) lowest-lying isomers, respectively. It possesses the optimized B–B distance of r_{B-B} = 1.73 Å between central B and its four closest B neighbours and La-B distance of $r_{\text{La-B}}$ = 2.68 Å between central B and four La atoms. The calculated HOMO-LUMO gap of ΔE_{gap} = 1.96 eV at PBE0 well supports the high chemical stability of the monoanionic complex. Its four equivalent La atoms possess the natural atomic charge of $q_{La} = +1.18 |e|$, the electronic configuration of La[Xe]5d^{1.60}6s^{0.14}, and the total Wiberg bond order of WBI_{La} = 3.14. The La-B coordination bond order between La and central B is calculated to be $WBI_{La-B} = 0.26$ which is slightly lower than the other La-B bond orders between La and its neighboring B₈ ring. Extensive molecular dynamics simulations indicate that $La_4B_{15}^{-}$ (4) is also highly dynamically stable at 800 K (Fig. S5a, ESI^{\dagger}). As the actinide analogue of La₄B₁₅⁻ (4), $D_2 \operatorname{Ac_4B_{15}}^{-}(4')$ also appears to be highly stable both thermodynamically and dynamically (Fig. S5b, ESI⁺). However, extensive global searches indicate that, unlike B₇ and B₈ rings, B₉ rings are too big in size to form stable conjoined double-ring metal complexes.

We further extend the complex series to the crown-like C_{3v} M_4B_{18} (5/5') which consist of three conjoined B_8 rings sharing a B₃ triangle on the top, with three equivalent octa-coordinate La atoms on the cage surface and one La atom inside at the center along the C3 molecular axis. Extensive GM searches indicate that La_4B_{18} (5) is the GM of the system at CCSD(T), with the second lowest-lying cage-like C_s isomer lying 0.07 eV higher in relative energy. Other cage-like isomers are found to be at least 0.81 eV less stable than the GM (Fig. S2(5), ESI⁺). La₄B₁₈ (5) possesses the optimized average La–B distances of $r_{\text{La–B}}$ = 2.89 Å between the central La atom and its B_{18} ligand and r_{La-B} = 2.72 Å between the octa-coordinate La atoms and their neighboring B₈ ligands on the surface, with the B-B distances of $r_{\rm B-B}$ = 1.69 Å within the three B-B dumb-bells shared by neighboring B_8 rings and r_{B-B} = 1.63 Å within the B_3 triangles on the top. The large HOMO-LUMO gap of ΔE_{gap} = 2.37 eV calculated for La4B18 (5) well supports the high chemical stability of the system with three conjoined B₈ rings. Similar situation exists in C_{3v} Ac₄B₁₈ (5'). In contrast, extensive efforts to construct boron frameworks with three conjoined B7 or B9 rings have proven to be unsuccessful. Furthermore, using the

B₈-based bottom-up approach established above from M_2B_8 (3/3'), $M_4B_{15}^-$ (4/4'), and M_4B_{18} (5/5'), the crown-shaped La_4B_{18} (5) with dangling bonds at the bottom can be further expanded to form boron frameworks with more than three conjoined B_8 rings based on the special topology of the B_8 ring. The perfect cubic $O_h M_7B_{24}^+$ (6/6') with six conjoined B_8 rings provide such special cases which can be viewed as two conjoining M_4B_{18} (5/5') units along the C_3 molecular axis.

Further increasing the cluster sizes, our group reported in two recent papers on singlet $O_h \operatorname{La}_7 \operatorname{B}_{24}^+$ (6) and doublet O_h La_7B_{24} with six conjoined B_8 rings, tetrahedral $T_d La_4B_{24}$ (8) with four conjoined B_9 rings, and tetrahedral $T_d \text{ La}_4 B_{29}^{-}$ (10) with four conjoined B9 rings encapsulating a tetrahedral BB4 core at the center.^{31,32} We focus here on their actinide analogues $O_{\rm h} \operatorname{Ac_7B_{24}^+}(6')$, $O_{\rm h} \operatorname{Ac_7B_{24}}$, $T_{\rm d} \operatorname{Ac_4B_{24}}(8')$, $T_{\rm d} \operatorname{Ac_4B_{29}^-}(10')$, and the newly obtained core-shell $C_1 M_5 B_{24}^+$ (9/9') (M = La, Ac). Both the endohedral singlet $Ac_7B_{24}^+$ (6') and doublet $O_h Ac_7B_{24}$ which can be viewed as six embedded La2B8 inverse sandwiches in a cube sharing a La vertex at the center possess perfect bodycentered structures with six equivalent octa-coordinate Ac atoms on the surface and one Ac atom at the center, $T_{\rm d}$ Ac_4B_{24} (8') as an ideal tetrahedral metallo-borospherene contains a boron framework consisting of four conjoined B9 rings with four equivalent nona-coordinate Ac atoms as integrated parts of the cage surface, $C_1 \operatorname{La}_5 \operatorname{B}_{24}^+$ (9) and $C_1 \operatorname{Ac}_5 \operatorname{B}_{24}^+$ (9') are slightly distorted endohedral tetrahedral metallo-borospherenes with four nona-coordinate metal atoms on the surface and one metal atom slightly off-centered, while $T_d \operatorname{Ac}_4 B_{29}^{-}$ (10') has a perfect tetrahedral core-shell structure with a tetrahedral $T_d BB_4$ unit encapsulated at the center. The low-lying isomers of the concerned species collectively depicted in Fig. S1-S3 (ESI⁺) indicate that lanthanide and actinide boride nanoclusters possess similar GMs and low-lying isomers in this size range.

Chemical bonding analyses

To better comprehend the high stability of La/Ac boride complexes with multiple conjoined B_n rings (n = 7-9), as examples, we performed detailed AdNDP bonding analyses on the closedshell $C_2 \text{La}_4 \text{B}_{13}^-(2)$, $D_2 \text{La}_4 \text{B}_{15}^-(4)$, and $C_{3v} \text{La}_4 \text{B}_{18}(5)$ to recover both the localized and delocalized bonds of the systems. As clearly indicated in Fig. 2a, $La_4B_{13}^{-}$ (2) possesses in the first row 5 2c-2e B-B periphery σ bonds on each of the two conjointed B_7 rings with the occupation number of ON = 1.87–1.89 |e| and 2 equivalent 3c-2e σ bonds on the two vertex-sharing B_3 triangles at the center with ON = 1.82 |e|, forming the σ -skeleton of the complex in an overall symmetry of C_2 . The second row contains 3 9c-2e $B_7(\sigma)$ -La₂($d_{\pi/\sigma}$) bonds over the $La_2B_7^-$ inverse sandwich on the left with ON = 1.90-2.00 |e| and 3 symmetrically distributed 9c-2e $B_7(\sigma)$ -La₂(d_{π/σ}) bonds over the La₂B₇⁻ inverse sandwich on the right, forming two equivalent local σ aromatic inverse sandwiches matching the 4n + 2 rule (n = 1) in La₄B₁₃⁻ (2). The two conjoined locally $\sigma\text{-aromatic systems in }La_4B_{13}^{-}$ (2) well correspond to the σ aromatic system in the experimentally observed inverse sandwich D_{7h} La₂B₇⁻ (1).²⁴ The 5c-2e π bond with ON = 1.67 in the third row and four 9c-2e $B_7(\pi)$ -La₂(d_{π}/σ) bonds with ON = 1.98–1.99 |e| at

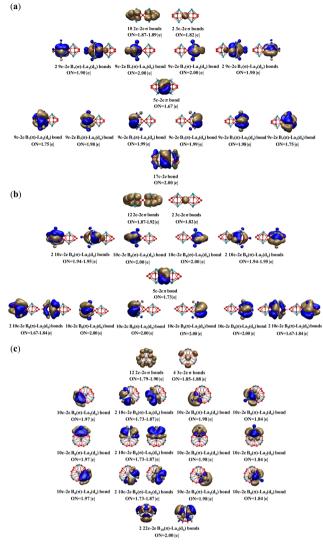


Fig. 2 AdNDP bonding patterns of the closed-shell (a) $C_2 \text{ La}_4\text{B}_{13}^{-}$ (2), (b) $D_2 \text{ La}_4\text{B}_{15}^{-}$ (4), and (c) $C_{3v} \text{ La}_4\text{B}_{18}$ (5), with the occupation numbers (ONs) indicated.

the center in the fourth row form a 10π aromatic system in $La_4B_{13}^-$ (2) matching the 4n + 2 rule with n = 2, rendering an overall π -aromaticity to the system. The remaining 2 9c-2e $B_7(\pi)$ – $La_2(d_{\delta})$ bonds on the left and right ends of the fourth row with ON = 1.75 |e| and the totally delocalized 17c-2e bond over the whole molecule with ON = 2.00 |e| in the fifth row help to further stabilize the system. Thus, the high stability of $La_4B_{13}^-$ (2) originates from the unique (d–p) σ , (d–p) π , and (d–p) δ coordination bonds between the four La atoms and the chiral C_2 B_{13} framework consisting of two conjoined B_7 rings.

As indicated in Fig. 2b, $La_4B_{15}^-$ (4) possesses a similar bonding pattern with $La_4B_{13}^-$ (2). It has 12 2c-2e B–B periphery σ bonds on the axially chiral B_{15} ligand and 2 equivalent 3c-2e σ bonds over the two vertex-sharing B_3 triangles at center in the first row in an overall symmetry of D_2 . There exist 3 10c-2e $B_8(\sigma)-La_2(d_{\pi/\sigma})$ bonds on the La_2B_8 inverse sandwich on the left and 3 10c-2e $B_8(\sigma)-La_2(d_{\pi/\sigma})$ bonds symmetrically distributed on the right, forming two equivalent locally σ aromatic systems matching the 4n + 2 rule (n = 1) in La₄B₁₃⁻ (2). Meanwhile, the 5c-2e π bond in the third row and four 10c-2e B₈(π)–La₂(d_{π/σ}) bonds in the fourth row at the center form a 10 π aromatic system following the 4n + 2 rule (n = 2). Different from La₄B₁₃⁻ (2), La₄B₁₅⁻ (4) with a sepcial electronic count possesses 2 10c-2e B₈(π)–La₂(d₈) bonds over the La₂B₈ inverse sandwich on the left and 2 10c-2e B₈(π)–La₂(d₈) bonds symmetrically distributed over the La₂B₈ inverse sandwich on the right in the fourth row. Again, it is the $\sigma + \pi$ dual aromaticity and unique (d–p) δ bonds between the four La atoms and the chiral D_2 B₁₅ framework with two conjoined B₈ rings that render high stability to La₄B₁₅⁻ (4).

The high-symmetry crown-like C_{3v} La₄B₁₈ (5) appears to have an even more interesting bonding pattern (Fig. 2c). It possesses 12 isolated 2c-2e σ bonds on the B₁₈ framemork and 4 isolated 3c-2e σ bonds on four B₃ triangles on the surface in the first row in an overall symmetry of C_{3v}. The following 15 10c-2e bonds in the second, third, and fourth rows are evenly distributed on three equivalent conjoined La2B8 inverse sandwiches embedded in the crown-shaped structure, including 3 10c-2e $B_8(\sigma)$ -La₂(d_{σ/π}) bonds, 1 10c-2e $B_8(\sigma)$ -La₂(d_{π}) bond, and 1 10c-2e $B_8(\pi)$ -La₂(d_{δ}) bond on each of the three conjoined La_2B_8 (3) inverse sandwiches, rendering local σ (4n + 2 = 6, n = 1) and π (4n + 2 = 2, n = 0) dual aromaticity to the system. The remaining 2 22c-2e bonds in the fifth row are delocalized over the whole neutral complex. Similar situations occur in Ac_4B_{18} (5'). The bonding patterns obtained above for $La_4B_{13}^{-}(2)$, $La_4B_{15}^{-}(4)$, and $La_4B_{18}(5)$ are well in line with their corresponding canonical molecular orbitals (CMOs) depicted in Fig. S6 (ESI[†]).

The 3D aromatic nature of D_{7h} La₂B₇⁻(1), C_2 La₄B₁₃⁻(2), D_{8h} $La_{2}B_{8}$ (3), $D_{2} La_{4}B_{15}^{-}$ (4), $C_{3v} La_{4}B_{18}$ (5), $D_{9h} La_{2}B_{9}^{-}$ (7), and C_{1} $La_5B_{24}^{+}$ (9) is further evidenced by their calculated nucleusindependent chemical shift (NICS) values. Based on the calculated NICS-ZZ components, Fig. 3 depicts their iso-chemical shielding surfaces (ICSS) with the Z-axis parallel to the designated C7, C2, C8, C2, C3, C9, or C1 molecular axes to illuminate the chemical shieldings in these complexes, compared with that of benzene D_{6h} C₆H₆. Obviously, the space inside the complexes or within about 1.0 Å above the metal atoms in vertical direction belong to chemical shielding regions with negative NICS-ZZ values (highlighted in yellow), while the chemical de-shielding areas with positive NICS values (highlighted in green) are located outside the B_n ligands in horizontal direction. The ICSSs of these complexes appear to be similar to that of the prototypical aromatic benzene C₆H₆.³³

1D, 2D, and 3D actinide boride nanomaterials

The highly stable cubic $O_h \operatorname{Ac_7B_{24}^+}(6')$ and $O_h \operatorname{Ac_7B_{24}}$ as embryos can be extended in *x*, *y*, and *z* three directions to form periodic actinide boride nanomaterials 1D $\operatorname{Ac_{10}B_{32}}(11')$ (*P*4/*mmm*), 2D $\operatorname{Ac_3B_{10}}(12')$ (*P*4/*mmm*), and 3D $\operatorname{AcB_6}(13')$ (*Pm3m*) with the optimized lattice parameters of a = 8.70 Å, a = b = 4.17 Å, and a = b = c = 4.22 Å at PBE, respectively. These results are well comparable with the corresponding values of a = 8.66 Å,

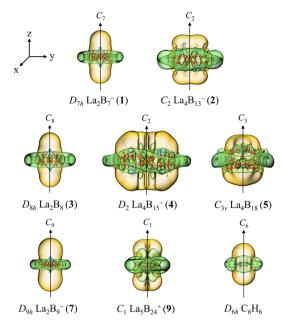


Fig. 3 Calculated iso-chemical shielding surfaces (ICSSs) of $La_2B_7^{-}$ (**1**), $La_4B_{13}^{-}$ (**2**), La_2B_8 (**3**), $La_4B_{15}^{-}$ (**4**), and La_4B_{18} (**5**), D_{9h} $La_2B_9^{-}$ (**7**), and C_1 $La_5B_{24}^{+}$ (**9**), in comparison with that of benzene D_{6h} C₆H₆. Yellow and green regions stand for chemical shielding and de-shielding areas, respectively.

a = b = 4.17 Å, and a = b = c = 4.16 Å calculated for their lanthanide boride counterparts 1D La₁₀B₃₂, 2D La₃B₁₀, and 3D LaB₆ at the same theoretical level.³² The calculated La–B distances of $r_{Ac-B} = 3.10$ Å for 3D AcB₆ (13'), $r_{Ac-B(x)} = r_{Ac-B(y)} =$ 3.08 Å, and $r_{Ac-B(z)} = 3.10$ Å for 2D Ac₃B₁₀ (12') also correspond well with the corresponding values in their lanthanide boride counterparts.³² Similar Ac–B distances are obtained for 1D Ac₁₀B₃₂ (11'). It is worth noticing that cubic B₂₄ frameworks with six conjoined B₈ rings in O_h Ac₇B₂₄⁺ (6') and O_h Ac₇B₂₄ are well maintained in its 1D, 2D, and 3D nanostructures with infinite conjoined B₈ rings in periodical arrangements (Fig. 4).

Fig. 5 depicts calculated band structures and projected density of states (PDOS) of the most-concerned 2D Ac_3B_{10} (12') and 3D AcB_6 (13') which are expected to be synthesized in experiments. As expected, these nanostructures all appear to be metallic in nature, similar to their lanthanide counterparts. Both the B 2p orbitals and Ac 6d orbitals contribute to the calculated densities of states near the Fermi level. For 2D Ac_3B_{10} (12'), Ac-6d orbitals make major contributions to the PDOS above the Fermi level, while B-2p orbitals dominate the PDOS below it. However, for 3D AcB_6 (13'), B-2p orbitals dominate in both situations, with Ac 6d orbitals making minor contributions.

The calculated elastic constants of $C_{11} = 456$ Gpa, $C_{12} = 29$ Gpa, and $C_{44} = 114$ Gpa, Young's modulus of Y = 340 Gpa, and Poisson's ratio of $\nu = 0.17$ for 3D AcB₆ (13') are well comparable with the corresponding measured⁵⁸/calculated values of $C_{11} = 463/468$, $C_{22} = 45/23$, and $C_{44} = 89/91$ Gpa, Y = 308/313 Gpa, and $\nu = 0.22/0.19$ obtained for the experimentally known 3D LaB₆ (13). The results obtained above indicate that,

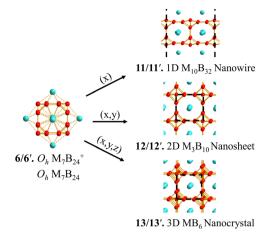


Fig. 4 Optimized structures of cubic $O_h M_7 B_{24}^+$ (**6**/**6**'), $O_h M_7 B_{24}$, 1D $M_{10}B_{32}$ nanowire (**11/11**'), 2D $M_3 B_{10}$ nanosheet (**12/12**'), and 3D $M B_6$ nanocrystal (**13/13**') at PBE level (M = La, Ac).

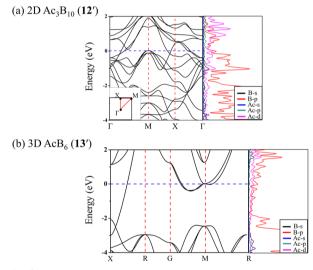


Fig. 5 Calculated band structures and projected densities of states (PDOS) of (a) 2D Ac $_3B_{10}$ (12') and (b) 3D AcB₆ (13') at PBE level.

similar to their lanthanide boride counterparts, low-dimensional actinide boride materials 1D $Ac_{10}B_{32}$ (11'), 2D $Ac_{3}B_{10}$ (12'), and 3D AcB_6 (13') are expected to have excellent electronic, mechanical, and thermal properties, in particular, AcB_6 (13') could be superhard materials stable at high temperatures.

Conclusions

In summary, we have predicted in this work a series of novel lanthanide boride $(La_4B_{13}^- (2), La_4B_{15}^- (4), La_4B_{18} (5), and La_5B_{24}^+ (9))$ and actinide boride $(Ac_2B_7^- (1'), Ac_4B_{13}^- (2'), Ac_2B_8 (3'), Ac_4B_{15}^- (4'), Ac_4B_{18} (5'), Ac_7B_{24}^+ (6'), Ac_2B_9^- (7'), Ac_4B_{24} (8'), Ac_5B_{24}^+ (9'), and Ac_4B_{29}^- (10'))$ nanoclusters based on boron frameworks composed of mono or conjoined B_n rings (n = 7-9). The newly proposed 1D $Ac_{10}B_{32} (11')$ nanowire, 2D $Ac_3B_{10} (12')$ nanosheet, and 3D $AcB_6 (13')$ nanocrystal based on

periodic boron frameworks composed of infinite conjoined B_8 rings expand the previously reported low-dimensional lanthanide boride nanomaterials to their actinide boride counterparts, presenting viable possibility to synthesize actinide boride nanomaterials with novel properties in macroscopic quantities.

Conflicts of interest

There are no conflicts to declare.

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